The Rietveld Method —
A Historical Perspective*

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Abstract
During the early sixties the least-squares method of adjusting structure parameters to minimise the differences between observed and calculated values was already well established in crystallography. From there it was, in retrospect, only a small step to refrain from using the integrated intensities as observed values but to use the actual measured profile intensities obtained by step scanning the powder diagram. The Rietveld method was first reported at the I.U.Cr. congress in Moscow in 1966. However, it was not until 1975, when it was also applied to X-ray diffraction, that it became widely accepted. Nowadays its use is no longer confined to elastic neutron powder diffraction, but to all diffraction techniques producing complex diffraction diagrams.

1. Introduction
As a Ph.D. student at the University of Western Australia between 1961 and 1964, I became thoroughly acquainted with X-ray and neutron diffraction techniques through experiments being conducted at the HIFAR in Lucas Heights (N.S.W.). The emphasis was on single crystal diffraction since, even then, the powder method was regarded as inferior, particularly for structure refinement. During that period the computer entered the scientific field and long, tedious structure factor and density calculations could be obtained more or less instantly. Using first the ‘Mercury’ computer at the University of Oxford and the ‘Silliac’ computer at the University of Sydney, and later the IBM 1620 at the Physics Department of the University of W.A., computers became an integral part of my crystallography work. They were, incidentally, also important for my later work.

2. Previous Efforts
After obtaining my Ph.D. degree in 1964 (Rietveld 1963), I joined the neutron diffraction group of the Reactor Centrum Nederland (now Netherlands Energy Research Foundation ECN). This group had only just been established and was principally engaged in the construction of a neutron powder diffractometer. The emphasis here was mainly on powder diffraction techniques, because it was apparent that no large single crystals could be grown of the materials that were then of

interest. The first crystal structures to be determined were rather simple and of a high symmetry, with the result that the peaks were well resolved and integrated intensities could easily be obtained for further refinement. However, with more complex compounds and of a lower symmetry, the overlap of peaks became so severe that separating them became practically impossible. In an effort to overcome this problem, the resolution of the diffractometer was significantly increased by using a wavelength of 2.6 Å and eliminating the higher order wavelengths using a filter of pyrolytic graphite (Loopstra 1966). This proved to be of appreciable value, especially for structure determination. For structure refinement, however, the increase of resolution certainly resulted in a better defined profile, but often not to such an extent that the peaks were completely resolved. The solution then was to refine the structure by using not only single intensities as data but also groups of overlapping intensities (Rietveld 1966a). This worked well, but the fact remained that all extra information contained in the profile of these overlapping peaks was lost. The following step was to separate the overlapping peaks by trying to fit gaussian peaks using least-squares procedures. This method also had its limitations, however, and did not work for severe overlap.

3. Use of Profile Intensities

Before the advent of computers, data reduction was a must in crystallography in order to be able to handle a relatively complex structure. Integrated intensities were therefore the smallest data elements one could practically work with. To consider using the individual intensities constituting a diffraction peak as data was completely unrealistic. With the experience of using computers for single crystal structure refinements and having seen their enormous capacity for handling large amounts of data, the spectre of increasing the number of data by a factor of ten by using the profile intensities instead of the integrated intensities constituted no real barrier. In the first refinement program, the profile intensities were corrected for background and were read in together with the value of the relative contributions each constituent peak made, i.e. the value of \( w_{i,k} \) in the expression \( y_i = \sum_k w_{i,k} S_k^2 \) (Rietveld 1967), where \( S \) is the structure factor. These values were calculated from the unit-cell dimensions and the wavelength, and zero-point and half-width values measured directly from the diagram. Also, for single peaks the integrated values were used rather than the profile intensities, because the gaussian peak shape did not fit that well at lower angles. Later a correction for this asymmetry was introduced. The non-refinement of the profile parameters can be explained by the fact that the computer then available, the Electrologica X1, was not powerful enough to solve a least-squares problem of more than a limited number of parameters. With the arrival of the larger Electrologica X8 computer, the program was written in its recent form and includes the capability of refining structure as well as profile parameters (Rietveld 1969b). Twenty-seven copies of this program, written in Algol (Rietveld 1969a) and later, in 1972, in FORTRAN IV, were distributed to institutes all over the world and this has greatly contributed to the acceptance of the method.

4. Acceptance of the Method

The method was first reported at the Seventh Congress of the I.U.Cr. in Moscow in 1966 (Rietveld 1966b). The response was very low, or rather non-existent, and it was not until the full implementation of the method was published (Rietveld 1969b)
that reactions came. At this time, the method was mainly used to refine structures obtained by fixed wavelength neutron diffraction; a total of 172 structures was solved in this way before 1977 (Cheetham and Taylor 1977). In the previously mentioned paper (Rietveld 1969b), it had been suggested that the method could also be applied to X-ray data, but it was not until 1977 with the paper of Malmros and Thomas (1977) that the method became generally accepted for neutron as well as X-ray powder diffraction, with fixed wavelength as well as fixed angle. This is reflected in an increasing number of citations to the original papers (Rietveld 1967; 1969b) as published in the Science Citation Index. Fig. 1 shows the number of citations between the years 1967 and 1986. Since the publication of these papers, many authors have contributed to an improvement in the method. The review article by Albinati and Willis (1982) gave a good impression of the state of the method at that moment. Many more papers on the method have appeared since, often with unexpected applications.

5. Conclusions

It has been most gratifying for me to experience how the Rietveld method has contributed to a renewed interest in powder diffraction techniques, even to the extent that in some applications it replaces single crystal techniques. The method is proven to be sound and has given results almost as good as single crystal data. The possible underestimation of the standard deviations as pointed out by Sakata and Cooper (1979) may serve as a reminder to all users that the method is not to be treated as a black box. One must be continually aware of the limitations, not only of this method, but in general of all least-squares methods. In this respect I fully agree with Prince (1981) who stated that: ‘If the fit of the assumed model is not adequate, the precision and accuracy of the parameters cannot be validly assessed by statistical methods.’
References


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